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Heterogeneous sorption of radionuclides predicted by crystal surface nanoroughness

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ABSTRACT

Reactive transport modeling (RTM) is an essential tool for the prediction of contaminants' behavior in the bio- and geosphere. However, RTM of sorption reactions is constrained by the surface site assessment. The reactive site density variability of the crystal surface nanotopography provides an “energetic landscape”, responsible for heterogeneous sorption efficiency, not covered in current RTM approaches. Here, we study the spatially heterogeneous sorption behavior of Eu(III), as an analogue to trivalent actinides, on a polycrystalline nanorough calcite surface and quantify the sorption efficiency as a function of surface nanoroughness. Based on experimental data from micro-focus time-resolved laser-induced luminescence spectroscopy (μ TRLFS), vertical scanning interferometry (VSI), and electron back-scattering diffraction (EBSD), we parameterize a surface complexation model (SCM) using surface nanotopography data. The validation of the quantitatively predicted spatial sorption heterogeneity suggests that retention reactions can be considerably influenced by nanotopographic surface features. Our study presents a way to implement heterogeneous surface reactivity into a predictive SCM for enhanced prediction of radionuclide retention.

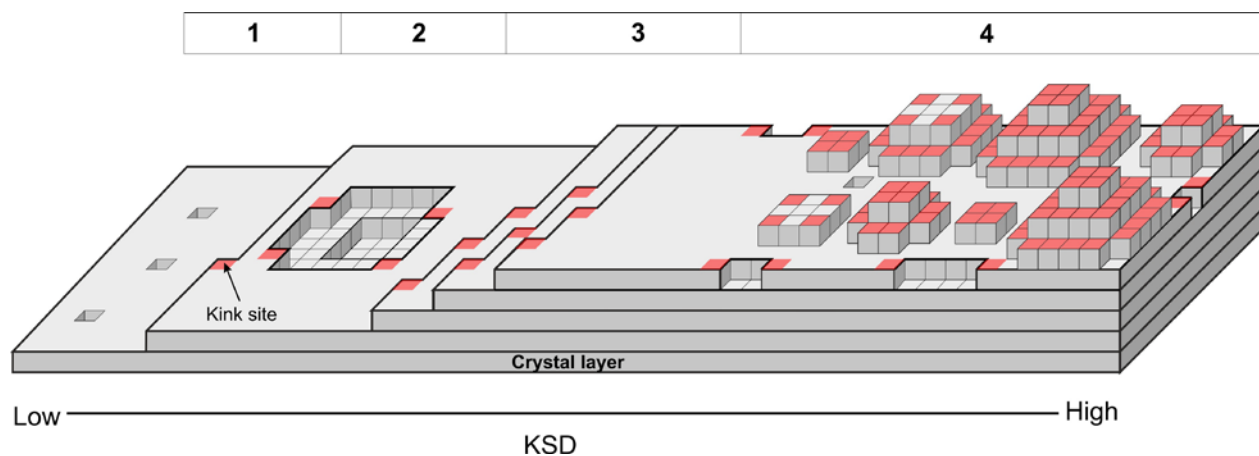
Keywords: Sorption reactions, Crystal surface reactivity, μ TRLFS, Surface complexation modeling, Radionuclide migration

1 INTRODUCTION

2 The potential migration of actinides constrains the safety assessment of disposal concepts for
3 nuclear waste in deep geological formations. Such concepts for underground storage facilities
4 require reliable predictions for 100,000s of years to diagnose potential migration of stored
5 radionuclides into the biosphere. Reactive transport models are crucial in providing such
6 predictions and the heterogeneity of physicochemical input values defines the reliability of the
7 numerical results. A key challenge is the upscaling of the molecular processes to quantify the
8 radionuclide retention processes, bridging the gap from laboratory scale experiments to large scale
9 facilities. Molecular mechanistic insights are provided experimentally and analytically by
10 spectroscopic techniques such as Time-Resolved Laser Fluorescence Spectroscopy (TRLFS)^{1, 2},
11 and Extended X-ray Absorption Fine Structure Spectroscopy (EXAFS)^{3, 4}, and numerically by
12 molecular dynamic simulations (MD), providing insights into the structure of the adsorbed ions at
13 specific surface sites^{5, 6}. Upscaling and implementing these insights into continuum scale reactive-
14 transport models is challenging because of the intrinsic variability of surface reactivity at the pore
15 scale. Typical RTM approaches employ uniform surface parameters such as specific surface area
16 (SSA) and site density, uniform electrostatic diffuse layer potentials, and simplified uniform
17 surface protonation models to simulate adsorption processes of radionuclides through various
18 surface complexation reactions implemented by surface complexation models (SCM)⁷⁻¹³ to yield a
19 simple retention coefficient in RTM at the continuum scale¹⁴⁻¹⁶. Consequently, simplifications such
20 as the assumption of macroscopically homogeneous behavior¹⁷ as well as surface area
21 normalizations¹⁸ are commonly applied. These simplifications limit our ability to model and predict
22 the consequences of varying surface reactivity.

23 Analytically, the recent development of the μ TRLFS technique allows us to identify interfacial
24 speciation with μ m-spatial resolution and the very high sensitivity required to investigate surface

25 complexes^{19, 20}. This technique revealed heterogeneous sorption behavior of the trivalent actinide
26 analog Eu(III) on a granite rock surface that cannot be quantitatively explained by mineral type
27 and composition alone¹⁹. Consequently, insight into the intrinsic variability of surface reactivity is
28 required to improve our predictive capabilities of reactive transport processes²¹. The surface
29 reactivity may depend on several factors, including crystal defects and surface nanotopography as
30 well as crystal orientation. From an energetic point of view, a dominant factor is the kink site
31 density (KSD) of the crystal surface.²² Figure 1 illustrates different nanotopographic configurations
32 of the crystal surface with specific kink site densities. With increasing KSD, these include (1)
33 atomically flat terraces with point defects, (2) etch pits walls with steps and kinks resulting from
34 dissolution processes, (3) surface steps with kinks along cleavage planes, and (4) very rough
35 surface portions with a multitude of edge and kink sites owing to mechanical treatment (polishing,
36 grinding, preparation of powdered sample material) of crystalline matter. Previous studies
37 suggested that increasing KSD results in enhanced surface reactivity for, e.g., sorption²³.



38
39 **Figure 1** Schematic representation of nanotopographic characteristics of a crystal surface, showing
40 (1) an atomically flat terrace with point defects, (2) an etch pit wall with steps and kinks, (3) surface
41 steps with kinks due to crystal cleavage, and (4) a very rough surface due to mechanical treatment
42 such as polishing. Kink sites are marked in the red color and the kink site density increases from
43 left to right in this scheme.

44 Here, we investigate the sorption of Eu(III) on calcite to experimentally exemplify the variability
45 of surface reactivity as a function of surface nanotopography of monomineralic but polycrystalline
46 collector surfaces during sorption processes. Eu(III) serves as a luminescent probe with excellent
47 spectroscopic properties²⁴ and a chemical analogue for the trivalent actinides, some of which
48 (Am(III), Pu(III)) determine the radiotoxicity of nuclear wastes over 100,000s of years. Calcite was
49 selected as one of the most widely distributed mineral phases in the earth's crust, with demonstrated
50 importance for the migration of radionuclides²⁵ and other contaminants in the geosphere²⁶, by both,
51 the complexation reactions at the mineral surface^{27, 28} as well as incorporation processes within the
52 crystal lattice.^{25, 29} Due to the presence of only a single mineral phase in this study, we can exclude
53 mineralogical effects and instead focus on the surface energy variability induced by crystallography
54 or topography. Our strategy is the quantitative mapping of surface sorption efficiency data,
55 including information about speciation and hydration of Eu(III) using μ TRLFS¹⁹. These data are
56 correlated with surface nanotopography maps from vertical scanning interferometry and crystal
57 orientation data from electron backscatter diffraction (EBSD) analysis. The spatially-resolved
58 information allows us to parameterize an improved SCM using surface nanoroughness data that
59 account for heterogeneities in crystal surface energy, and which makes it possible to predict
60 heterogeneous sorption behavior. The modeling results were validated using experimental data
61 from μ TRLFS.

62 **METHODS**

63 **Experimental setup.** Two calcite samples were utilized to investigate the effect of the different
64 nanotopographical endmembers on the sorption efficiency of Eu(III) on the crystal surface. One
65 sample was a polished calcite crystal with a high density of edge and kink sites due to its rough
66 surfaces (manufacturer: KORTH Kristalle GmbH, Altenholz, polished along (10-14), the other is
67 a cleaved calcite surface (calcite island spar, Chihuahua, Mexico, WARD's 46E1438) with varying

68 degrees of step densities analyzed by VSI. The two samples were exposed to 10^{-6} M aqueous
69 [^{152}Eu] $\text{Eu}(\text{NO}_3)_3$ in 0.1 M NaCl solution [at pH 7.5 by immersing their surface upside down in the
70 solution for 15 minutes. No alterations of the surface topography are detected using interferometry
71 microscopy surface analysis. The sorption was quantified via autoradiographic imaging using a
72 BAS-IP SR storage phosphor screen that was read out with a pixel size of 25 μm after exposure
73 using an Amersham Typhoon Biomolecular Imager (GE Lifesciences). Absolute concentrations
74 were yielded by calibration of the image in ImageJ using a series of europium standards of known
75 concentration.

76 As a second step, the investigations were then expanded to a more realistic system with respect to
77 calcite-bearing host rocks. Another sorption experiment was performed using polycrystalline
78 calcite. A cylindrical drilled marble sample (diameter: 1.5 cm, length: 1 cm, Großsoelk, Austria,
79 composition: calcite (95 wt. %), quartz (< 2 wt. %), muscovite (< 1 wt. %), phlogopite (< 1 wt. %)
80 and ore minerals (<1 wt. %) ³⁰, was polished using an oil-based mono-crystalline diamond
81 suspension (crystal size: 50 nm) and partially masked with a layer of 550 nm of gold ³¹ as a height
82 reference for surface analysis at the nanometer scale. To identify the crystallographic orientation
83 of the crystal grains in the sample surface a grid of 200 x 150 points (resolution: 10 μm) was
84 analyzed with EBSD using a Zeiss NVision 40 scanning electron microscope equipped with a
85 Bruker eFlash^{Hr}- detector and a QUANTAX CrystAlign EBSD-System (acceleration voltages:
86 10kV, beam current: 2.5nA). ³² First, to provide a surface roughness variability similar to what is
87 expected in natural systems after diagenetic reactions, a dissolution experiment was performed by
88 reacting the sample surface with air-equilibrated, aqueous 2.2 mM Na_2CO_3 at pH 8.8 for 1 hour
89 using a custom-made flow-through cell (cell volume: 300 μL , flowrate: 30 mL/h). The
90 experimental conditions are identical to those described in the literature for studying dissolution
91 kinetics of calcite surfaces at far-from-equilibrium conditions ²². Second, a batch europium sorption

92 experiment was conducted by immersing the sample surface into a solution of 10^{-5} mol/L $\text{Eu}(\text{NO}_3)_3$
93 and 0.1 mol/L NaCl with a pH value of 8.5 for 24 hours. In this step, the larger concentration and
94 longer sorption period were made necessary to enable the μTRLFS studies^{19, 20}. The sample was
95 then rinsed with the background solution and dried. Spatially resolved information on the europium
96 sorption efficiency and the adsorbed species was gained using μTRLFS mapping of a $0.5 \text{ mm} \times$
97 1.5 mm subsection with a spatial resolution of $20 \text{ }\mu\text{m}$.

98 The surface topographies of all the samples were analyzed using a Sensofar S neox white-light
99 vertical scanning interferometer equipped with Nikon DI interferometry objectives for $20 \times$, $50 \times$,
100 and $100 \times$ magnifications (see SI, Table S1 for detailed specifications).

101 **μTRLFS .** μTRLFS measurements were conducted with a setup designed in-house as described in
102 Molodtsov et al. 2019¹⁹. The laser beam was provided by a pumped (Surelite SL I-20 @ 355 nm,
103 Continuum) dye laser (NarrowScan @ Exalite 389/398 1:1 mix, Radiant Dyes). The pulse energy
104 was set to approximately $20 \text{ }\mu\text{J}$ to prevent possible laser ablation. The wavelength was fixed at 394
105 nm to maximize the light absorption of Eu^{3+} (${}^7\text{F}_0 \rightarrow {}^5\text{L}_6$ transition) resulting in higher sensitivity
106 for luminescence detection²⁴. The laser beam was focused by a 50x long-distance objective
107 (MLWD-50X, Newport) to a spot with approximately $20 \text{ }\mu\text{m}$ in diameter. The sample surface was
108 aligned to the focal plane of the laser beam with a z-stage (KT-RS60, Zaber) and then scanned in
109 a $20 \text{ }\mu\text{m}$ grid with an x-/y-stage (M-423 driven by TRB25CC and controlled with CONEX-CC,
110 Newport). The gathered luminescence light was redirected to the detector (Shamrock SR303i
111 spectrograph combined with DH320T-18U-63 iCCD camera, Andor) to collect the full spectral
112 information in each data point. Spectra were measured with 100 accumulations over 5 seconds in
113 total with a gate width and exposure time of the iCCD of 10 ms. To evaluate the data a python-
114 based software (picroTRLFS EVAL, HZDR) was used. The spectra were analyzed with respect to
115 the peak areas of the ${}^5\text{D}_0 \rightarrow {}^7\text{F}_1$ (${}^7\text{F}_1$) and ${}^5\text{D}_0 \rightarrow {}^7\text{F}_2$ (${}^7\text{F}_2$) transitions and their fluorescence decays

116 after applying a linear background correction. The fluorescence intensity (sum of peaks ${}^7F_{1+} + {}^7F_2$)
117 is correlated to the number of Eu(III) atoms and therefore corresponds to the sorption efficiency.
118 The sorption efficiency of each pixel was calculated by normalizing the peak sum by the highest
119 measured luminescence signal. The luminescence peak ratio (F_2/ F_1) corresponds to the sorption
120 strength and also serves as a distinguishing indicator for speciation because the F_2 peak is a
121 hypersensitive electric multipole transition, and its intensity reacts strongly to changes in Eu(III)'s
122 chemical surroundings, while the F_1 peak as a magnetic dipole transition is mainly unaffected.²⁴
123 Lifetimes of the luminescence decays were recorded in several selected data points. The
124 luminescence decay patterns could be reconstructed by bi-exponential decay functions with a fixed
125 long lifetime of 3600 μ s, which represents the well-known lifetime of the Eu^{3+} incorporation
126 species into the crystal lattice of calcite.²⁵ This fixation is useful because the long-lived species
127 usually have a quite low luminescence intensity, which is therefore much more affected by
128 background correction uncertainties. As a result, the short lifetime can then be determined much
129 more reliably. In some cases, the observed luminescence decay pattern was described by a mono-
130 exponential decay function for residual-minimization purposes. The resulting lifetimes were then
131 correlated to the number of remaining water molecules $n(\text{H}_2\text{O})$ in the first coordination sphere of
132 Eu^{3+} by the empirical equation^{33,34}:

$$133 \quad n(\text{H}_2\text{O}) \pm 0.5 = \frac{1.07}{\tau} - 0.62, \quad (1)$$

134 where τ denotes the measured luminescence lifetime (in a unit of ms). The value of n can range
135 from zero (incorporation) up to nine remaining water molecules (fully hydrated Eu^{3+} aquo ion).
136 Intermediate amounts of water molecules left correspond to inner sphere complexation of the Eu^{3+}
137 ion, here sorption to the calcite surface.

138 **Sorption data analysis.** Areas of different quantities of sorption were identified in the
139 autoradiogram of the polished and cleaved samples and representative high-resolution topography

140 images were recorded. The sorption data on the polycrystalline calcite material from μ TRLFS with
141 a pixel size of 20 μm were cross-referenced with different grains (A-J) identified by EBSD. The
142 grains were ordered with increasing europium sorption (see Figure S1) and Student t-tests were
143 performed to analyze statistically significant differences in europium sorption ($p=0.05$). The data
144 from neighboring grains with no statistically significant differences were combined into single
145 datasets. This created four combined datasets with statistically significant differences in europium
146 sorption efficiency, AF, IJHG, EDC, and B (in order of increasing europium sorption, see Table
147 S2 and S3). Of the four datasets the grains with the most available data points – A, I, D, and B (in
148 order of increasing europium sorption) – were chosen for a more detailed correlative analysis of
149 the europium sorption and surface nanoroughness. VSI maps of each grain were measured and Sq
150 values, the root mean square deviation or the standard deviation of the height data, of 20 $\mu\text{m} \times 20$
151 μm pixels were extracted from the images and correlated with the μ TRLFS sorption data.
152 For a more detailed investigation, the 20 $\mu\text{m} \times 20 \mu\text{m}$ pixels were each divided into subpixels with
153 a smaller field of view (FOV) down to a maximum of 64 subpixels with a minimum FOV of $2.5 \times$
154 $2.5 \mu\text{m}^2$ which were analyzed according to their Sq value. The applied minimum FOV size of
155 topographic data utilized for the calculation of the Sq parameter is limited by the spatial resolution
156 and consecutive bisection algorithm used. The resulting datasets of Sq values were then averaged
157 for every original pixel to mitigate the influence of larger structures such as scratches and holes in
158 the crystal surface on the Sq-data (note that averaging the Sq values of the subpixels and measuring
159 one Sq value for the whole pixel will give different results, with a higher influence of extreme
160 values with the latter approach). These converged datasets were then correlated with the europium
161 sorption efficiency. In a separate analysis different Sq ranges were investigated for their correlation
162 with the sorption efficiency by only averaging the subpixels with Sq values in the ranges of 0-5
163 nm, 5-10 nm, 10-50 nm, and >50 nm (see Figure S2).

164 **Numerical model.** According to the lifetime analysis, we utilized the surface complexation
165 reactions to describe Eu^{3+} adsorption on the calcite surface in a short time experiment. Therefore,
166 the SCM was developed using the aqueous speciation reactions and surface complexation reactions
167 in Table S4. The SCM is an electrostatic double-layer model^{11,13} in which the net surface charge
168 balances the net charge in the diffuse layer⁷. The identical site density and specific surface area
169 (SSA) are 5 sites/nm² and 0.262 m²/g, respectively²⁸. The chemical reactions were calculated using
170 PHREEQC v3.5⁷ with the Nagra/PSI Chemical Thermodynamic Database³⁵. The SCM results were
171 validated using measured sorption efficiency data from Zavarin et al. 2005²⁸ (Figure S3 in SI).

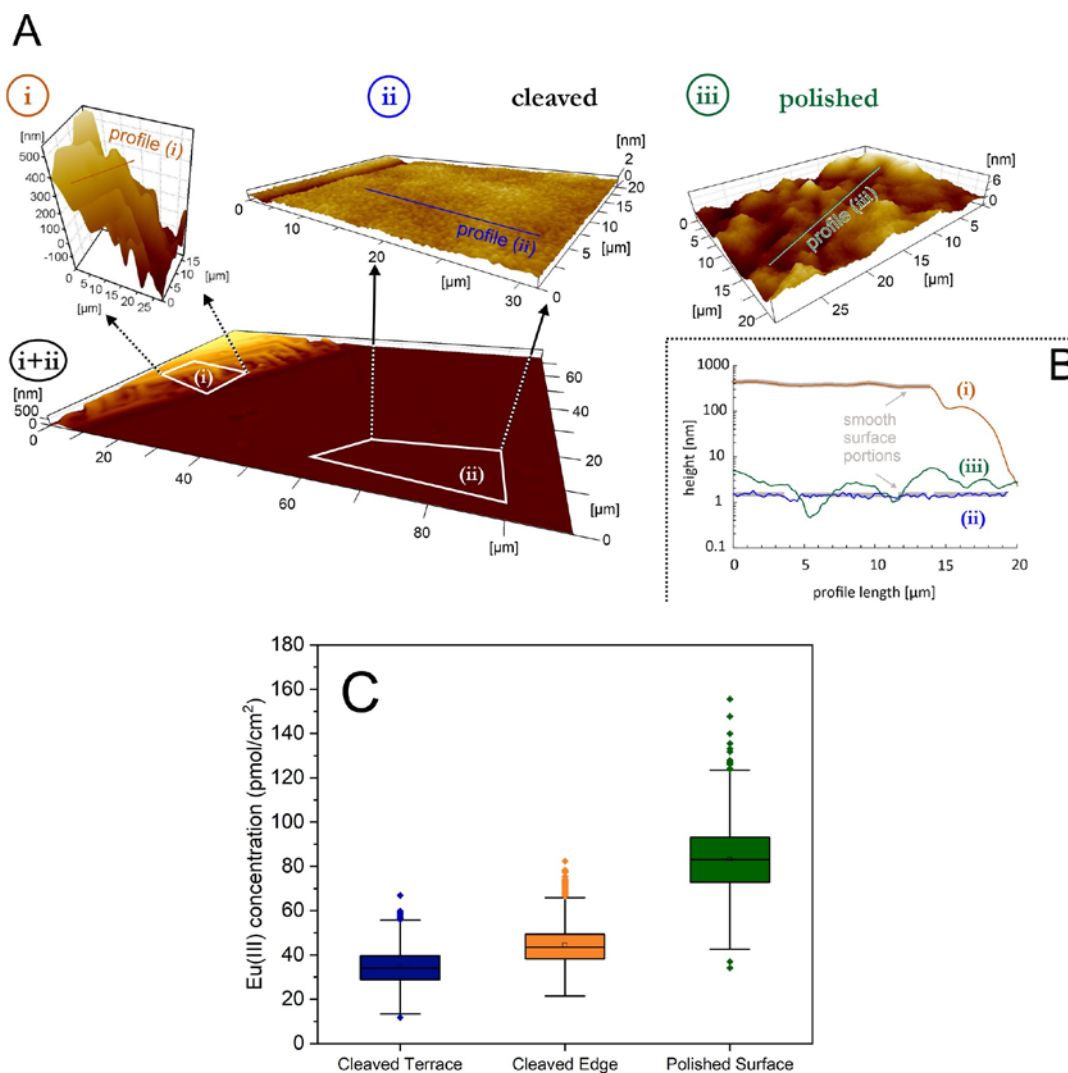
172 **Parametrization of the predictive model.** The topographical data for the predictive model is
173 obtained from phase shifting interferometry (PSI) measurement with a high spatial resolution of 46
174 nm × 46 nm. The Sq value of each pixel is calculated at a 0.46 × 0.46 μm² FOV using 10 × 10
175 pixels of the surface topography data. Since the averaged Sq value is sensitive to the FOV, the
176 calibration curve for the predictive modeling was adapted for the higher resolution. Based on the
177 experimentally validated correlation of the optimized slope α with the FOV of the Sq value for ROI
178 I and II (Figure S4 in SI), α was set to 0.03 for the predictive modeling. See below for a detailed
179 discussion of these procedures. The use of a slope not specifically optimized for the surface in
180 question may cause deviations in the prediction of the sorption efficiency, however, the predicted
181 spatial sorption patterns and their trends with the evolution of the surface are valid.

182 **RESULTS AND DISCUSSION**

183 **Calcite nanotopography effects on Eu(III) sorption.** The influence of surface topography on
184 sorption efficiency was studied using calcite crystal samples with different topographies that
185 allowed us to isolate the effects of specific surface features. The use of single-crystal samples
186 allowed us to study specific structural endmembers, i.e. very low KSD to very high KSD, even
187 though the exact KSDs are not accessible to measurement. Figure 2 shows the surface topographies

188 of a cleaved and a polished calcite crystal (I) and the absolute concentrations of sorbed Eu(III) in
189 a cleaved edge area (A), a cleaved terrace (B), and a fine-polished surface (C) (II). The polished
190 crystal surface (C, Figure 2I) exhibits a highly nanorough surface that consists of a high amount of
191 kink sites due to the fine polishing (cf. Figure 1 surface 4) producing a surface that may be linked
192 to a powder sample. On the surface of the cleaved sample, two distinct structural features can be
193 observed: distinct steps (A) that contain various degrees of steps according to the height difference
194 (cf. Figure 1 surface 3) and flat terraces (B) in between the steps that can be made up by the large
195 flat areas with a low amount of step density and point defects (cf. Figure 1 surface 1). Figure 2II
196 shows the different concentrations of sorbed europium that the surface features A, B, and C lead
197 to, as quantified by autoradiography. The highest amount of sorbed Eu(III) can be observed on the
198 polished sample with a highly nanorough surface while the cleaved terrace with an approximately
199 flat surface exhibits the lowest one, with the edge structure falling in between.

200 The observed heterogeneous sorption behavior suggests a variability of surface reactivity that
201 reflects a heterogeneous distribution of crystal surface energy that can be quantified by surface
202 topography analysis. The surface energy strongly depends on the number of surface building blocks
203 such as kink sites and step edges¹⁷. A previous study²² investigated heterogeneous surface reactivity
204 and the probability of variations in kink site density. Additional literature data^{36, 37} confirm the
205 importance of the surface kink sites as the dominating intrinsic factor of enhanced surface reactivity
206 during, e.g., dissolution and precipitation reactions. We thus hypothesize that such surface energy
207 distribution affects the sorption efficiency, as suggested by kinetic Monte Carlo simulations (KMC)
208 at the atomic scale³⁸. Moreover, a molecular dynamics (MD) study investigating the reactivity of
209 the calcite-water interface shows the number of surface reactive sites for adsorbing water molecules
210 is higher at rough surface positions with kink sites and step edges compared to the flat surface
211 portions with no surface roughness⁵.



212

213 **Figure 2.** (A) surface topographies of cleaved calcite surface ((i) and (ii)) and polished calcite

214 surface (iii). (i) shows a stepped edge due to cleaving with an overall height of about 500 nm. (ii)

215 shows a flat cleavage plane with a maximum height of ~ 2 nm due to a single scratch along the

216 otherwise flat surface. Note the different exaggerations of the z scale compared to the x and y scale

217 of the maps (i-iii). (B) shows height distributions of the profile lines (i), (ii), and (iii). (C) Box-

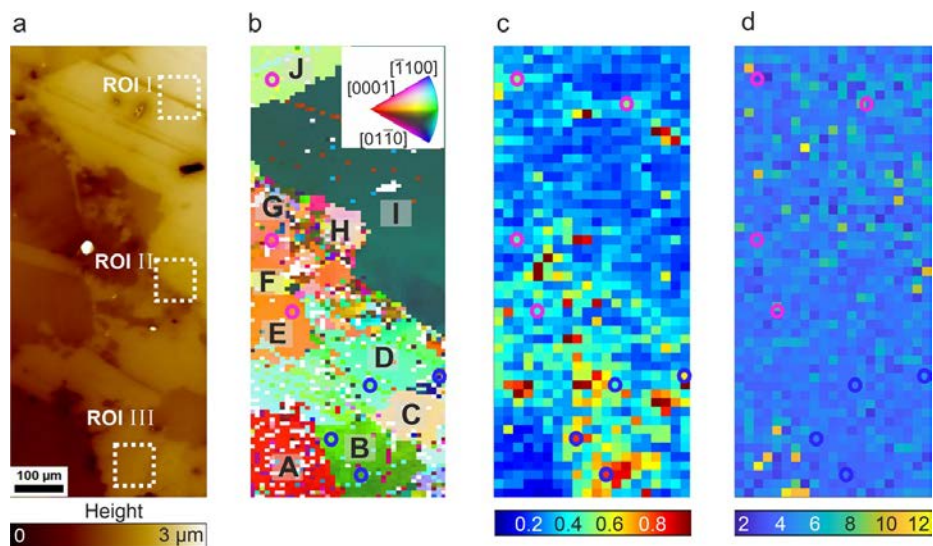
218 Whisker plots of the measured absolute concentrations of sorbed Eu(III) among the areas (i), (ii),

219 and (iii). The quantitative concentration trend indicates the effect of crystal surface nanotopography

220 (cf. Fig. 1) on sorption efficiency.

221 **Heterogeneous sorption of Eu(III) on polycrystalline calcite.** Now, we investigate the surface
222 of polycrystalline calcite towards variability in sorption efficiency. Figure 3 shows an overview of
223 the investigated calcite crystal surface maps including (a) the surface topography, (b) crystal
224 orientations of the polycrystalline calcite material, (c) normalized sorption efficiency including
225 selected spots where fluorescence lifetimes were collected, and (d) speciation distribution. The
226 surface is composed of multiple crystal grains with diameters of about 10 μm to several 100 μm
227 with varying crystal orientation. The sorption efficiency was determined based on the total
228 luminescence intensity of Eu(III) as described in the methods section. The luminescence intensity
229 reveals a heterogeneous distribution of the Eu(III) sorption, even within single crystal faces,
230 varying by a factor of around 10 between lowest and highest concentrations. Several grains show
231 statistically significant differences in sorption efficiency, however, no correlation between crystal
232 orientation and sorption efficiency is found (see below and SI). An analysis of the ratio of the 5D_0
233 $\rightarrow {}^7F_2$ and 7F_1 luminescence transitions and lifetimes of the excited states provides information on
234 the speciation of Eu(III) on the sample surface, as they are influenced by the bond strength²⁴ and
235 the number of water molecules³³ in the first coordination sphere of the excited europium (see SI
236 for detailed analysis of luminescence lifetime and speciation). In contrast to the heterogeneously
237 distributed sorption efficiency, the mapping of the ${}^7F_2/{}^7F_1$ peak ratio which is correlated to
238 speciation (Figure 3d) shows a mostly homogeneous distribution. Most pixels exhibit peak ratios
239 ~ 5 , indicating that the dominant speciation of Eu(III) is surface sorption or incorporation species.
240 A few pixels show a peak ratio of seven or above, but only where the luminescence intensity is
241 very low, which leads to larger uncertainties in the determination of the peak ratios.
242 The measured lifetimes at selected surface sites (see Figure 3c, circles) are shown in the SI (Table
243 S5). We generally find two distinct lifetimes: A short lifetime, which correlates to ~ 2 H_2O in the
244 coordination sphere of Eu(III) and a longer lifetime which indicates a complete loss of hydration.

245 The short lifetime in combination with the typical peak ratio discussed above can be interpreted as
 246 an inner-sphere surface sorption complex that has been previously reported^{25, 39}. This species is
 247 only present where the luminescence intensity is high and can be considered to be the major fraction
 248 of Eu on the mineral surface. The long lifetime and thus complete loss of hydration suggest
 249 incorporation into the calcite lattice²⁵. The species appears homogeneously distributed over the
 250 whole surface, albeit always only in low quantities. As Eu is a common impurity in natural
 251 calcites⁴⁰, and because it seems unlikely that incorporation occurred in the short duration of our
 252 experiments²⁹ we interpret this species as Eu(III) incorporated into the natural calcite before our
 253 experiments. We can then conclude that as a first approximation the Eu(III) sorption efficiency is
 254 heterogeneously distributed over the surface, but the dominant species is the same inner sphere
 255 complex throughout.

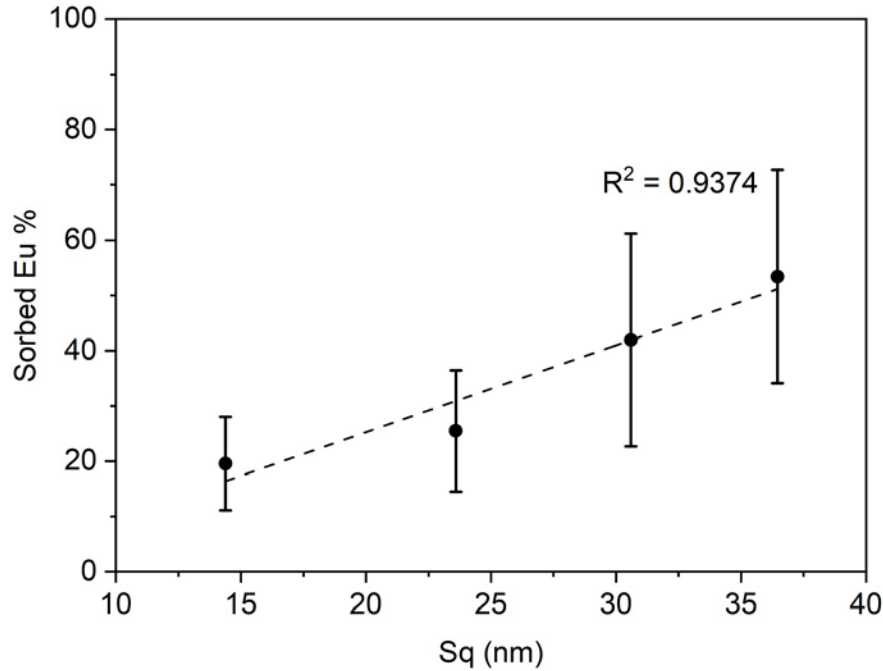


256
 257 **Figure 3.** Correlative microscopy of (a) Surface height map, field-of-view (FOV) size = 1 mm ×
 258 0.4 mm; ROIs I-III refer to three representative sub-regions for model validation (see Figure 6).
 259 (b) Crystal orientation map based on EBSD data, different colors represent specific crystal
 260 orientations, see the legend in Figure S1 for details. (c) Normalized sorption efficiency (normalized
 261 $F_1 + F_2$ μTRLFS peak sum) with lifetime data points marked with a hollow dot in blue (high

262 intensity) and purple (low intensity), spatial resolution: $20\ \mu\text{m} \times 20\ \mu\text{m}$. **d**, Peak ratio F_2/F_1 ,
263 indicating speciation/bonding strength, spatial resolution: $20\ \mu\text{m} \times 20\ \mu\text{m}$.

264 **Surface nanoroughness parametrization replaces the SSA parametrization towards**
265 **improved surface complexation modeling.** The two datasets of the cleaved and polished surface
266 illustrate the influence of the surface nanotopography is as a major factor for the measured Eu(III)
267 sorption variability (Fig. 2). At the nanometer scale, the spatial configuration surface building
268 blocks defines the surface reactivity and, thus, the sorption efficiency²³. Nanoroughness can be
269 considered a proxy to quantify the occurrence of such surface building blocks. In order to quantify
270 surface roughness, we select the Sq parameter, the root-mean-square (r.m.s.) deviation of the
271 surface height⁴¹ calculated from the surface topography data, to test its suitability to explain the
272 observed heterogeneous sorption efficiency. Parameter Sq quantifies the height variability more
273 reliably⁴² than the arithmetic average roughness Sa. Additionally, we tested the potential use of the
274 surface area ratio parameter, S_{dr} , as a proxy parameter, see Figure S6 in SI.

275 For the polycrystalline calcite sample, we identify four datasets of different grains, AF, IJHG, EDC,
276 and B by μTRLFS with statistically significantly different Eu(III) sorption (see Table S2 & S3).
277 The crystal orientation within these datasets varies widely with no discernable trend (see Figure
278 S1). No correlation between crystal orientation and Eu(III) sorption efficiency has been found.
279 Furthermore, no correlation between the sorption efficiency and the absolute sample height exists
280 as regions of distinctly different absolute surface heights show similarly low sorption efficiencies
281 (see Figure 3a and c). Grain-specific data (grains: A, B, D, and I) of the sorption efficiency of
282 Eu(III) versus the average Sq roughness of the grain surfaces show a linear correlation ($R^2 = 0.94$),
283 cf. Figure. 4, proving the Sq parameter as a suitable proxy for surface reactivity. Note that effects
284 of crystal orientation may result in a variability of surface nanoroughness, encompassed in the
285 roughness parameter.



287

288 **Figure. 4.** Normalized amount of europium sorption efficiency based on μ TRLFS data collected

289 on grains A, I, D, and B as a function of the Sq roughness of the grain surface.

290 For a more detailed site-specific investigation as well as for testing the potential use of

291 topographical features for parametrization of the numerical approach, the sorption efficiencies of

292 all pixels of the four different grains with statistically significant different Eu(III) sorption

293 efficiencies were correlated with the Sq value of the field-of-view of each pixel (Figure 5a). Each

294 pixel covers an area of $20 \mu\text{m} \times 20 \mu\text{m}$ (in accordance with the Eu sorption data resolution), which

295 is equivalent to 200×200 pixels of the surface topography data shown in Figure 3a. Based on these

296 115 data points (Figure 5a), only a weak correlation ($R^2 = 0.16$) between Eu sorption efficiency

297 and Sq value can be identified. Overall, the data seems to be distributed between two regimes, i.e.,

298 one regime shows a larger increase of Eu sorption with elevated Sq values up to 40 nm (red color

299 area in Figure 5a), the other regime shows a strong effect of Sq on Eu sorption, mostly covering

300 low includes elevated Sq values up to 80 nm with minor or no correlation with Eu sorption

301 efficiency (blue color area in Figure 5a). A closer look at the VSI images reveals that in the latter
302 case, the Sq values are strongly influenced by larger surface structures, such as scratches or holes.
303 These features have a large impact on the Sq value but do not appear to significantly impact Eu(III)
304 sorption. Thus, to deconvolve the two regimes (large structures vs. nanoscale roughness) we
305 gradually reduced the field of view for the Sq value calculation down to a minimum of 2.5×2.5
306 μm^2 using consecutive bisection to divide each pixel into a maximum of 64 subpixels, resulting in
307 64 Sq values for each original pixel. Averaging these sub-pixel Sq values for each original pixel
308 results in a new Sq value for each pixel at a $2.5 \times 2.5 \mu\text{m}^2$ FOV, and the impact of the larger
309 structures is reduced due to the averaging procedure. The mitigation of the influence of the larger
310 structures is warranted by a more detailed analysis of the correlation between sorption efficiency
311 and Sq values. A separate analysis of subpixels in the ranges of $< 5 \text{ nm}$, $5\text{-}10 \text{ nm}$, $10\text{-}50 \text{ nm}$, and $>$
312 50 nm was performed. These ranges were chosen characterize different surface building blocks:
313 The range of $0\text{-}10 \text{ nm}$ reflects the critical nanoroughness of the surface, the intermediate range is
314 typical for smaller holes left by the EBSD process and dissolution etch pits ($10\text{-}50 \text{ nm}$), and the
315 range $> 50 \text{ nm}$ corresponds to large scratches and holes due to the sample preparation. A correlation
316 of the chosen Sq ranges with the sorption efficiency shows correlations only for the first three
317 ranges, i.e., $0 - 50 \text{ nm}$. No correlation was found for Sq values above 50 nm , justifying the de-
318 emphasisation of larger structures by the averaging procedure in the previously described (see
319 Figure S2). Mechanistically, this indicates that specific nm-sized surface building blocks such as
320 steps and kinks are the main drivers of the increased sorption efficiency.

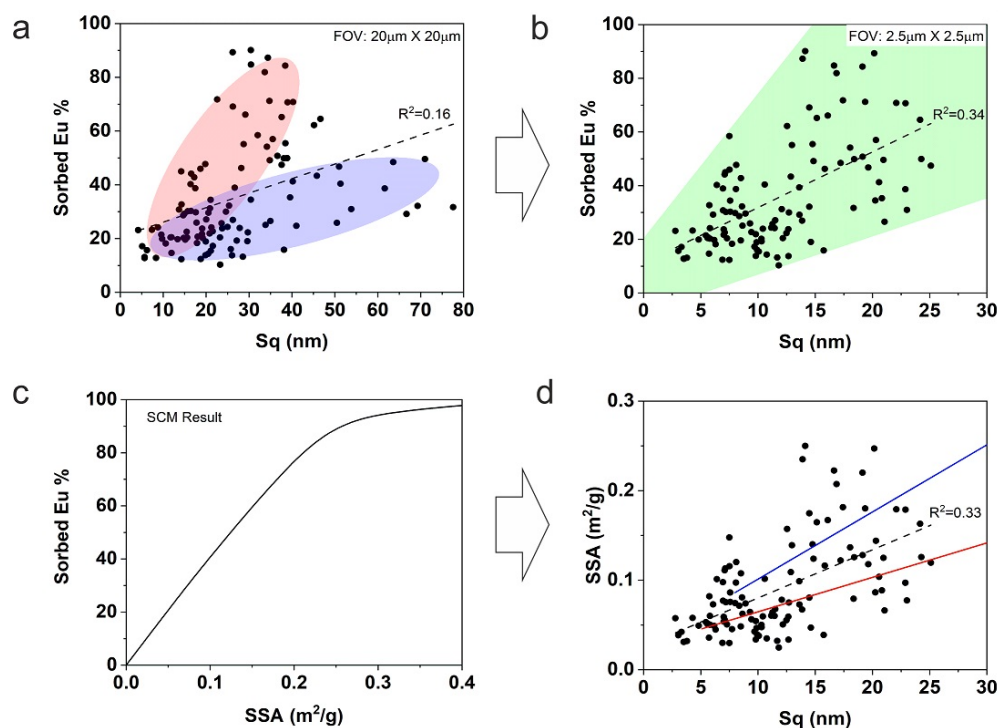
321 Using the newly calculated Sq values for each pixel, we see an increase of the coefficient of
322 determination (R^2) (see SI Figure S5), resulting in a linear correlation at a FOV of $2.5 \times 2.5 \mu\text{m}^2$
323 with $R^2 = 0.34$ (Figure 5b) corresponding to a Pearson's correlation coefficient r of 0.58. While
324 this may appear low, we like to point out that an r of 0.58 is considered to reflect a rather strong

325 effect for datasets with a large influence of other randomly distributed influences⁴³. The motivation
326 of our investigations is to test the predictive power of Sq as a proxy parameter in a polycrystalline
327 system that inherently reflects a combination of several variables. The important prediction of the
328 safety of a nuclear waste repository needs model parameter validation in a realistic system to avoid
329 oversimplification. The data showing a visible spread, nevertheless, the point cloud envelopes a
330 cone of possible linear relationships between Sq and Eu sorption efficiency. This general
331 correlation is potentially and locally affected by several other factors that cause a decreased
332 coefficient of determination for the overall pixel-by-pixel correlation when compared to the
333 averaged grainwise correlation. Influencing factors may include the crystal orientation, the
334 contribution of intrinsic Eu to the μ TRLFS signal, the still present impact of larger structures on
335 the Sq value, and effects of unspecific Eu(III) precipitation and calcite dissolution during the
336 sorption experiment. This leads to slightly different valid slopes in different areas of the sample.
337 Furthermore, assuming slightly increased Eu sorption signals due to the presence of incorporated
338 natural Eu and slightly exaggerated Sq values due to large topographic features for the outliers
339 would shift the highest Eu sorption data and highest Sq values to lower values. This would result
340 in a tendency of narrowing the spread of the cone of possible linear relationships (see Figure S7
341 for illustration). Possible corrections and improvements will be the subject of future investigations.
342 For this study, we choose the linear regression of all data points as the most parsimonious solution
343 to establish a correlation between sorption efficiency and surface nanoroughness. However, while
344 these results provide a quantitative link between surface nanoroughness and sorption efficiency,
345 they cannot be implemented into standard SCM approaches directly. Instead, these models rely on
346 the specific surface area (SSA) and site density (SSD) to quantify the available surface sorption
347 sites interacting with aqueous species at the solid/liquid surface. Typically, SCMs include a
348 description of the electric double layer (e.g. the Gouy-Chapman model⁴⁴) and conventional mass

349 action laws for intrinsic chemical reactions at the surface,⁴⁵ but adsorption is ultimately treated as
350 an interaction of metal ions with functional groups uniformly distributed on the surface.
351 Consequently, heterogeneous sorption variabilities as observed in our experiment and previously
352 reported in our previous work^{19,20} cannot be modeled by this standard approach. These variabilities
353 are based on heterogeneously distributed affinities of cations for distinct positions on the calcite
354 surface⁴⁶, which are interpreted to be related to the variable amount of coordinated water molecules
355 and Ca-O bond lengths⁴⁷ on the different positions of the crystal surface. These underlying affinity
356 variations hold true beyond our specific system, for example, an investigation of the interaction
357 between CO and a Cu surface demonstrated the CO-Cu binding energies at surface defects like
358 steps or kinks to be systematically larger than at the terraces⁴⁸. In general, cations preferably form
359 surface complexes with the typically undercoordinated functional groups at the position of surface
360 defects⁴⁹.

361 In surface complexation modeling, the SSD reflects the total surface sites per unit surface area
362 (nm^2), and the SSA reflects the total surface area per unit mass of mineral, which is typically
363 measured using the BET method⁵⁰. The BET-measured surface area shows a dependence on the
364 surface topography of the solid surface, which is linearly correlated with a surface roughness factor
365 , i.e., the ratio of the measured (rough) surface area to the projected (smooth) area⁵¹. In order to
366 implement spatial variability of the total number of surface sites based on variations of SSD and/or
367 SSA owing to surface roughness variability into an SCM, one could either modify the SSA and fix
368 SSD or vice-versa or modify both at the same time. We choose to vary the SSA and keep the SSD
369 constant, supported by a strong linear correlation of the interpolated surface area and the Sq value
370 (see Figure S8 in SI). As we have established a correlation between surface nanoroughness Sq and
371 sorption efficiency, linking the Sq of our sample to its SSA should enable us to implement the
372 observed interfacial heterogeneity into a predictive model. An SCM was set up (see SI for details)

373 to establish a quantitative relationship between the Eu sorption efficiency and SSA (Figure 5c).
374 Ultimately, we aim to link our experimental data to literature data using surface data collected by,
375 e.g., BET adsorption techniques. Thus, the concept utilizes a modification of the SSA parameter
376 based on the bulk literature data as a starting point. An opposite choice of varying SSD and keeping
377 SSA constant would yield identical results (see Figure S9) as the process is mathematically
378 equivalent. This allows us to quantitatively link SSA and Sq pixel by pixel (Figure 5d) by cross-
379 referencing the experimental sorption efficiency data (Figure 5b) and the model result (Figure 5c).
380 The quantitative trends in Figs. 5b and d exhibit consistent trends with the investigations of both
381 KMC³⁸ and MD⁵ approaches, which confirms the validity of using surface nanoroughness data as
382 a proxy for the intrinsic crystal surface energy, which has a significant impact on sorption
383 efficiency. Therefore, we utilize this linear trend between SSA and Sq to parameterize a predictive
384 SCM with the assumption that the SSA has a monotonically increasing linear relationship with Sq
385 values.

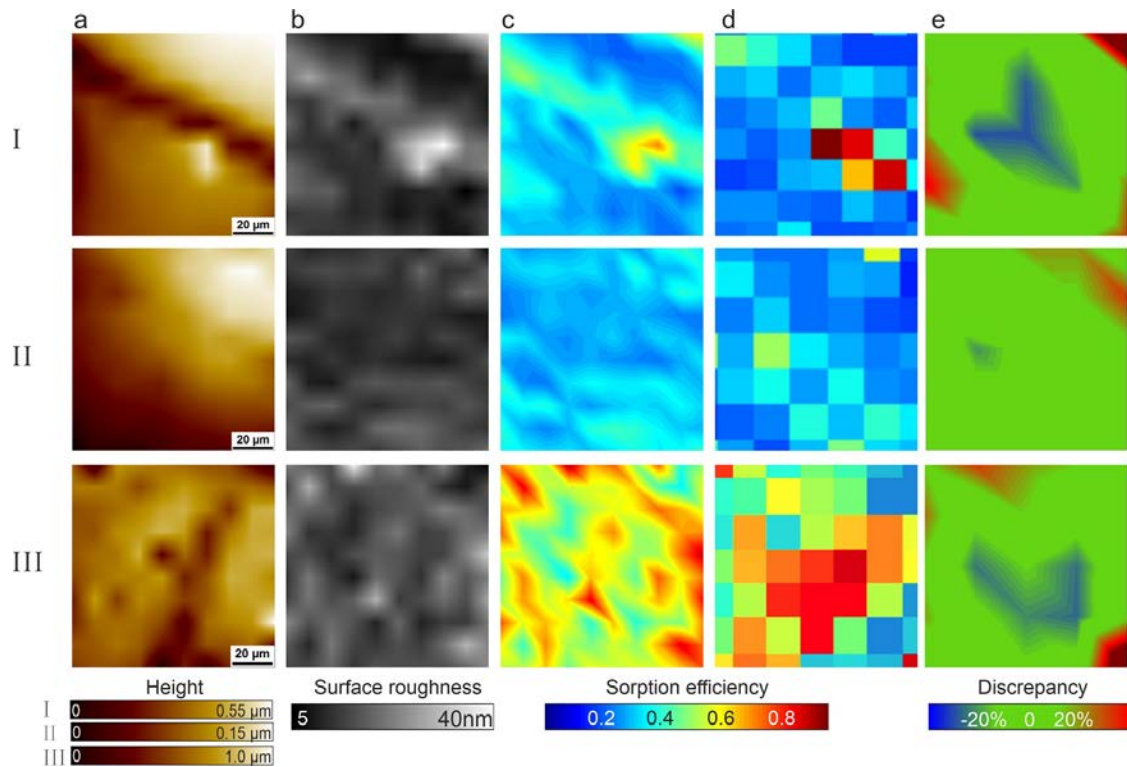


386

387 **Figure 5.** Quantitative correlation between SSA (model) and Sq (experiment) based on
388 experimental data analysis and SCM results. **a,** The scatter plots of measured sorption efficiency
389 versus Sq at a $20 \times 20 \mu\text{m}^2$ FOV. **b,** The scatter plots of measured sorption efficiency versus Sq at
390 $2.5 \times 2.5 \mu\text{m}^2$ FOV (right). **c,** SCM results of sorption efficiency versus SSA based on the reactions
391 and thermodynamic data in Table S4. **d,** A scatter plot of SSA versus the averaged Sq at a $2.5 \times$
392 $2.5 \mu\text{m}^2$ FOV. The SSA of each point is calculated by the sorption efficiency through the
393 quantitative curve in Figure 2c. The correlation between SSA and Sq shows a linear curve (gray
394 dashed line) with a coefficient of determination, $R^2 = 0.33$. The red solid line and blue solid line
395 represent the calibration curves used to parameterize the SCM for model validation in ROI I, II,
396 and III, respectively. See text for detailed discussion.

397 **Improved SCM parametrization and model validation.** The quantitative relationship between
398 Sq and SSA derived in the previous section is now used to parameterize a spatially resolved SCM
399 using experimentally determined nanoroughness data. Based on the quantitative analysis in Figure
400 2, we derive the slope of the linear function of SSA vs. Sq as α , with $\alpha = 0.0054$. The ROIs I, II,
401 and III in Figure 1a are selected as representative regions with available VSI-measured surface
402 topography and sorption data from μTRLFS for model validation. Based on the surface topography
403 of the ROIs (Figure 6a), the corresponding surface roughness maps (Figure 6b) were calculated
404 with a pixel resolution of $10 \mu\text{m} \times 10 \mu\text{m}$ by averaging the Sq value of $2.5 \mu\text{m} \times 2.5 \mu\text{m}$ sub-pixels.
405 We then use the quantitative relationship between Sq and SSA as reflected in α to calculate the
406 SSA of each $10 \mu\text{m} \times 10 \mu\text{m}$ pixel in the three ROIs and, from that, the Eu sorption (Figure 6c).
407 Naturally, the SCM results of the SSA parameterization are very sensitive to the value of α and the
408 spread of the cone of possible linear relationships is wide. Consequently, we performed a sensitivity
409 analysis to optimize the value of α for every ROI. Optimized α values were obtained for the
410 individual ROIs by maximizing the fit of the modeled Eu sorption in each ROI with the

411 corresponding experimental values (Figure S10). This yielded different calibration curves for each
412 ROI which are plotted in Figure 5d and fall well within the spread of the experimental data, with
413 ROI III exhibiting a steeper slope (blue line with $\alpha = 0.0075$) than the other two (red line with $\alpha =$
414 0.0038). The SSA modification via Sq parametrization allows us to model similar sorption trends
415 in all three cases. In general, a quantitative comparison of model results (Figure 6c) with the
416 measured data (Figure 6d) shows good agreement within a 20% discrepancy between model and
417 experiment (Figure 6e). In particular, the model underestimates the highest Eu sorption values in
418 ROI III, which may be caused by Eu impurities of the natural calcite crystals, as indicated by the
419 fluorescence lifetime measurement at this point of the sample (see SI Table S5, point 12/48). In
420 contrast to typical SCM approaches that employ a simple uniform SSA value^{9, 10}, this modified
421 SSA based on Sq analysis allows for the investigation of spatially resolved retention variability
422 based on the variability of crystal surface energy¹⁷. Choosing the right α will be critical in the
423 feasibility of this approach for predictive modeling and more research needs to be devoted towards
424 the clarification of the varying influences on the relationship between surface topography and
425 surface energy variation.



426
 427 **Figure 6.** SCM validations in ROIs I, II, and III. **a**, surface topography, spatial resolution = $10 \mu\text{m}$
 428 $\times 10 \mu\text{m}$; **b**, averaged surface roughness (S_q) at a $2.5 \mu\text{m} \times 2.5 \mu\text{m}$ FOV, spatial resolution = 10
 429 $\mu\text{m} \times 10 \mu\text{m}$; **c**, SCM results of sorption efficiency, spatial resolution = $10 \mu\text{m} \times 10 \mu\text{m}$; **d**,
 430 normalized experimental results of sorption efficiency measured using μTRLFS techniques, spatial
 431 resolution = $20 \mu\text{m} \times 20 \mu\text{m}$; **e**, the ratio of model vs. experimental sorption efficiency. Green color
 432 represents a range of $\pm 20\%$ discrepancy, red color: overestimation, blue color: underestimation.

433 **Predicting radionuclide sorption evolution.** As a next step, we explore the potential of our
 434 improved parameterization approach to predict heterogeneous sorption on the crystal surface of an
 435 independent set of experimental data. Figure 7 shows two surface topography maps, measured
 436 using phase shifting interferometry (PSI) of a calcite single crystal after 120 minutes (a I) and 135
 437 minutes (a II) of etching, respectively, representing the crystal surface evolution during dissolution.
 438 From the topographical data (Figure 7a), we again derived the S_q values, here with a spatial
 439 resolution of $0.46 \mu\text{m} \times 0.46 \mu\text{m}$ (Figure 7b), use S_q to derive the SSA with the previously

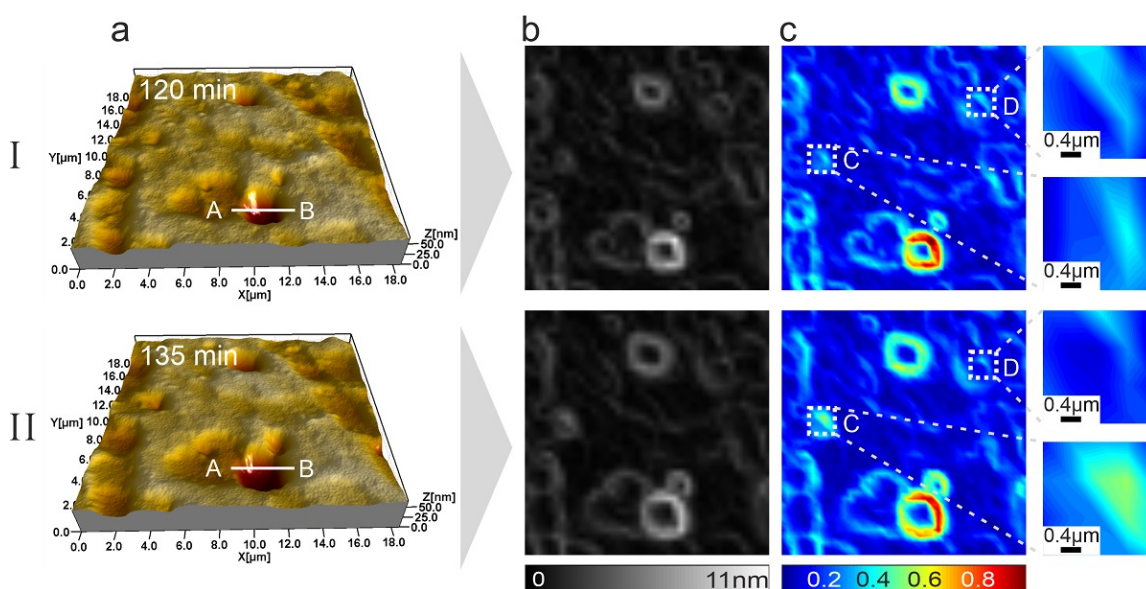
440 determined value for α , adapted for the higher resolution (see Figure S4), and predict heterogeneous
441 Eu(III) sorption (Figure 7c) based on our novel SCM parameterization approach.

442 A first important finding from this prediction is that Eu(III) is preferably adsorbed at surface steps
443 which exhibit higher nanoroughness and kink site densities. In contrast, flat terraces and nearly flat
444 bottoms of the surface pits are predicted to show lower adsorption. For the widening of etch pits
445 on the surface, such as in cross-section A-B in Figure 7, the areas of high surface roughness move
446 outwards with the evolution of the etched pit, resulting in a redistribution of the preferred sorption
447 areas. Another finding with important implications for the evolution of the overall surface
448 efficiency is that the sorption efficiency varies with the evolution of surface topography beyond
449 just a spatial redistribution of the preferred sorption sites. As an example, ROI C shows an overall
450 increase of the S_q value due to the etching which goes along with a 160% increase in sorption
451 efficiency. This enhanced sorption efficiency as a result of the evolving surface structure may be
452 linked to an increase in surface steps and kinks¹⁷ by etching. In contrast to the enhanced sorption
453 efficiency in ROI C, the sorption efficiency of ROI D exhibits a 25% reduction, linked to a reduced
454 S_q due to the removal of surface features by etching. Such behavior is explained by locally
455 contrasting surface step arrangement on a reacting crystal surface. One situation is a constant
456 number and density of single height steps on the crystal surface. The other situation is the formation
457 of step bands due to surface step bunching, resulting in a locally enhanced step density⁵².

458 The overall sorption efficiency of the entire domain and its evolution over time depends on the
459 balance between such subareas. Thus, SSA parameterization via surface nanoroughness in SCM
460 analysis may provide a tool to predict the evolution of radionuclide sorption caused by the changes
461 in surface building blocks. Consequently, the detailed quantitative insight into retention variability
462 gained by this approach may provide a more accurate description of sorption at the pore scale
463 contributing to an enhanced predictive capability on the core scale and beyond. An experimental

464 validation of such high-resolution modeling is not yet possible using current analytical approaches
 465 with limited spatial resolution. Nevertheless, the mechanistic background of specific crystal surface
 466 portions showing enhanced kink site densities (Figure 1) and, thus, elevated sorption uptake (Figure
 467 2) provides an opportunity for the validation of such modeling approaches. Consequently, the
 468 numerical results predict potential pattern evolution of reacting surfaces and offer opportunities for
 469 applications.

470



471

472 **Figure 7.** Prediction of the evolution of sorption efficiency for two high spatial resolution data,
 473 based on PSI techniques reflecting evolving surfaces at 120 min and 135 min dissolution reaction
 474 time. Surface area = $18.5 \mu\text{m} \times 18.5 \mu\text{m}$. **a**, surface topography, spatial resolution = $0.046 \mu\text{m} \times$
 475 $0.046 \mu\text{m}$. **b**, surface roughness (Sq), spatial resolution = $0.46 \mu\text{m} \times 0.46 \mu\text{m}$. **c**, SCM results of
 476 sorption efficiency, spatial resolution = $0.46 \mu\text{m} \times 0.46 \mu\text{m}$. ROI A-B shows the sorption
 477 preference at surface steps rather than the flat terrace. ROIs C and D show the enhanced and
 478 reduced sorption efficiency due to the changes in surface topography by etching, respectively.

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